Synthesis, Structure, and Magnetic Properties of a Novel Mixed-Valent Strontium Rhodium Oxide

John Bleddyn Claridge and Hans-Conrad zur Loye*

Department of Chemistry and Biochemistry, University of South Carolina, Columbia, South Carolina 29208

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Low-dimensional materials, both one- and twodimensional, have long been of interest to chemists and physicists because of their distinctive electronic and magnetic properties. $1-3$ Much work has been done on the halide perovskites with the BaNiO₃ structure⁴ and, recently, oxides with the $K_4CdCl_6^5$ structure, such as Sr3NiIrO6, have been shown to exhibit interesting magnetic properties associated with their highly anisotropic structures. $6-11$ Darriet has shown that these structures are closely related¹² and, furthermore, that it should be possible to form phases with structures intermediate between those of the BaNiO₃ and K_4CdCl_6 structure types with the general formula $A_{3n+3}A'_{n}$ $B_{n+3}O_{9+6n}$ ($n = 0$, BaNiO₃; n = ∞ , Sr₃NiIrO₆). So far, however, comparatively few oxides of this series with structures other than those of the end-members have been synthesized.13-²² Moreover, of these structures,

 * To whom correspondence should be addressed. E-mail: hanno@ psc.sc.edu.

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Figure 1. Observed (cross) and calculated (solid line) X-ray diffraction patterns of $Sr_6Rh_5O_{15}$. Tick marks indicate the positions of allowed Bragg reflections. The difference line, observed minus calculated, is located at the bottom of the figure.

the electronic properties have been reported for only one of the $n = 1$ members; $Ba_6Ni_5O_{15}$ is a Curie-Weiss paramagnetic insulator at all temperatures measured.13

In this paper we report on the synthesis, structural characterization, and preliminary analysis of the magnetic properties of $Sr_6Rh_5O_{15}$, an $n = 1$ member of the series A3*ⁿ*+3A′*n*B*ⁿ*+3O9+6*n*.

Polycrystalline $Sr_6Rh_5O_{15}$ was synthesized via a solidstate reaction. $SrCO₃$ (Alfa, 99.95%) and Rh metal (Engelhard, 99.95%) were intimately mixed and thoroughly ground, followed by pressing the powder into pellets and placing them into alumina crucibles. The pellets were initially heated in air at 850 °C for 10 h and then at 1150 °C for 9 days with intermittent grinding. The X-ray powder diffraction data were collected on a Rigaku D\Max-2200 powder X-ray diffractometer with Bragg-Brentano geometry using Cu $K\alpha$ radiation. The step-scan covered the angular range 10°-135° 2*^θ* in steps of 0.02° 2*θ*.

Structure refinements of $Sr₆Rh₅O₁₅$ were carried out in the space group *R*32 (No. 155) using the Rietveld method implemented in the computer programs GSAS²³ and RIETAN.^{24,25} Both programs yielded similar re-

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^a An overall thermal parameter *Q* (0.16(2) Å2) was used in the refinement. *Q* is defined as an isotropic thermal (*B*) applied to each individual atom in the structure.

Figure 2. Structure of $Sr_6Rh_5O_{15}$ (a) viewed along the *c*-axis and (b) along [110]. Sr cations are white circles; \overline{R} hO₆ polydedra are shaded gray.

sults, and only those using RIETAN will be discussed. *^R*3 (No. 146), *^R*3h (No. 148), *^R*3*^m* (No. 160), and *^R*3h*^m* (No. 166) are also consistent with the observed extinctions. Therefore, refinements were also carried out in *R*3, *R*3, and *R*3*m*. Both *R*3 and *R*3*m* gave significantly worse refinements than *R*32 (*R*3*m* is a supergroup of $R\bar{3}$ and *R*3*m* and was thus not refined). *R*3 gave a marginally better refinement, (*R*wp 10.81 vs *R*wp 10.96); however, the number of parameters was greatly increased and the atoms remained very close to the positions for *R*32, making it a less satisfactory solution. Consequently, we report the structure in *R*32. Estimated standard deviations were calculated using the method of Scott,26 which yields larger values than the conventional method, though more physically realistic. The profile of the diffraction peaks of $Sr_6Rh_5O_{15}$ was described by a pseudo-Voigt function. An overall thermal parameter and March-Dollase preferred orientation parameter were refined. Refinement of the peak asymmetry was allowed, and the background was described by a polynomial function with eight refineable coefficients.

Details of the Rietveld refinement of $Sr_6Rh_5O_{15}$ are given in Table 1. The best agreement obtained between the calculated and the observed profiles for $Sr_6Rh_5O_{15}$ is shown in Figure 1. The atomic positions for $Sr₆ Rh₅O₁₅$ can be found in Table 2. $Sr₆Rh₅O₁₅$ is closely related to the known compounds $Sr_6Co_5O_{15}^{14}$ and Ba₆- $Ni₅O₁₅;¹³$ however, to account for all the observed diffraction peaks, the unit cell was transformed $(a' =$ $-2a$, $b' = -2b$). Attempts to index the extra peaks on the basis of the incommensurately modulated composite structure observed for several related oxides^{15,28} were unsuccessful. The rhombohedral structure consists of two types of crystallographically distinct infinite onedimensional chains of tetramers composed of facesharing RhO_6 octahedra linked by RhO_6 trigonal prisms; these infinite chains themselves are separated by chains of Sr cations. The structure is shown in Figure 2. The differences between the $Sr_6Rh_5O_{15}$ structure and the $Sr_6 Co₅O₁₅$ structure appear to be due to a distortion of the strontium chains. Comparisons of the heavy atom positions between the average structure solved in the $Sr₆Co₅O₁₅$ cell and the larger cell give only small shifts $(0.07 Å) for all the atoms except Sr(3), which shifts$ 0.36(4) Å toward the trigonal prisms centered by Rh- (5). This is presumably caused by alternations in Rh-^O bond distances due to charge ordering. However, given the limitations of the current data set, collected on a laboratory X-ray powder diffractometer, there exist

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integrated intensities, respectively, *yi* is the weight derived from an error propagation scheme during the least squares refinement process, and $I_{K}(\text{obs})$ and $I_{K}(\text{calc})$ are the extracted and calculated Bragg peak intensities.

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Figure 3. (a) Temperature dependence of the magnetic susceptibility measured at 5 kG and (b) a χ T plot measured at 5 \hat{k} G for $\check{S}r_6Rh_5O_{15}$.

sufficiently large uncertainties in the oxygen positions as to preclude assignment of the oxidation states of the rhodium atoms based upon their metal oxygen bond lengths. Further studies using synchrotron XRD and/ or neutron diffraction in order to locate the oxygen atoms precisely would undoubtedly provide more structural information.

The magnetic susceptibility of $Sr₆Rh₅O₁₅$ was measured using a Quantum Design MPMS XL SQUID magnetometer. The temperature dependencies of the susceptibilities were measured under both field-cooled (FC) and zero-field-cooled (ZFC) conditions by heating the samples from 2 to 300 K in applied fields of 5 and 40 kG. The small diamagnetic contribution of the sample holder was corrected for by subtraction of a blank run. Figure 3a shows the magnetic susceptibility for $Sr₆Rh₅O₁₅$ measured at 5 kG both FC and ZFC, and Figure 3b shows a plot of χT versus *T* for the ZFC data collected in the 5 kG field. Between 300 and 150 K,

the system follows the Curie-Weiss law. The measured Curie-Weiss parameters, $C = 0.055(1)$ emu K/mol (Rh), θ = 28(2) K, and μ_{eff} = 0.66(3) μ_{B} , correspond to one unpaired electron per cluster (tetramer of octahedra plus trigonal prismatic site) containing five rhodium cations (spin only value of $0.66\mu_B$). One assignment of Rh^{III} and Rh^{IV} to trigonal prismatic and octahedral sites in the chains that is consistent with the magnetic data is as follows: Rh^{III} occupies all the trigonal prismatic sites and one of the octahedral sites, while Rh^{IV} occupies the remaining three octahedral sites. An average of only one unpaired electron per tetramer can arise due to metal-metal interactions, either Rh-O-Rh or direct Rh-Rh interactions within the tetramer, consistent with the very short Rh-Rh distances (particularly Rh1- Rh1 and Rh2-Rh2). Similar metal-metal interactions have been suggested for the structurally closely related BaNiO₃.^{29,30} These tetramers are weakly ferromagnetically coupled *intra*chain; the existence of this ferromagnetic coupling is supported by the positive Weiss constant and the increase in the magnetic moment (χT) with decreasing temperature, as seen in Figure 3b. There is an antiferromagnetic transition at 7 K followed by a precipitous drop in susceptibility below 7 K. The drop in susceptibility is greater than that predicted by most magnetic models for a randomly oriented powder, where the susceptibility is expected to drop to $\frac{2}{3}$ of χ_{max} . Similar drops in the susceptibility have been observed for several related chain type oxides such as $\rm Ca_3Co_2O_6^{11}$ and $Sr_3NiIrO_6^{8,31}$ and in cluster systems. However, in the chain type oxides the systems are magnetically frustrated, whereas in the cluster systems the transitions are broad. For $Sr_6Rh_5O_{15}$, the transition is relatively sharp and the FC and ZFC measurements are virtually identical, indicating that the spins are not magnetically frustrated below 7 K. Conductivity measurements, carried out on a pressed pellet between 300 and 15 K, support the conclusion that the material is a semiconductor/insulator.

In conclusion, we have synthesized $Sr_6Rh_5O_{15}$ which crystallizes in a novel superstructure of the $Sr_6Co_5O_{15}$ type. The compound is a paramagnetic semiconductor above 7 K, below which temperature an antiferromagnetic transition occurs.

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Supporting Information Available: Listings of bond distances, Rietan list file, and calculated and observed peak positions (11 pages). Ordering information is given on any current masthead page.

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